

3^{ème} reunion plénières de la Fédération Hydrogène (FRH2) 22-26/05/2023

Multiple phase transitions in $Y_{1-y}R_yFe_2$ hydrides and deuterides

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Outline

1. Interest of AB_2 Laves phase compounds
2. YFe_2 structure and influence of H absorption
3. Methods
4. Results
 - Structures of YFe_2D_x compounds
 - Phase transitions
 - Thermodesorption
 - Order-disorder transitions
 - Magnetic transitions
 - Magnetism and magnetic entropy in $Y_{1-x}R_xFe_2$ compounds
5. Conclusions



Wood museum, Taiwan

1. Interest of AB_2 Laves phase compounds

➤ A large variety of AB_2 Laves phases: > 1000 compounds

➤ A great number of applications:

F. Stein, A. Leineweber, Laves phases: a review of their functional and structural applications and an improved fundamental understanding of stability and properties, *J. Mater. Sci.*, 56 (2021) 5321-5427.

- RFe_2 : Giant magnetostriction for actuators, sonars ...

➤ Hydrogen storage application :

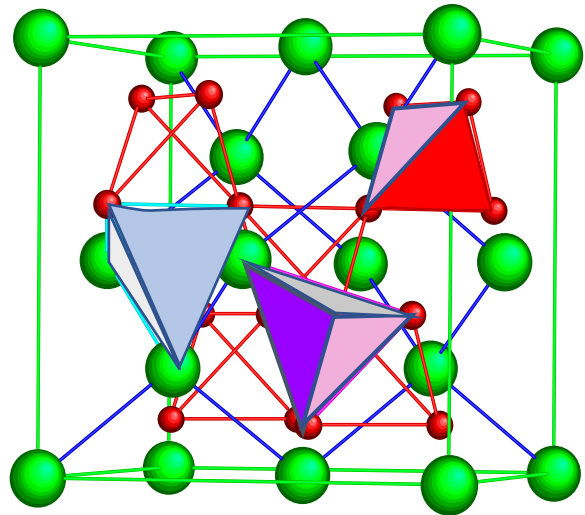
V.A. Yartys, M.V. Lototskyy, Laves type intermetallic compounds as hydrogen storage materials: A review, *J. Alloys Compds*, 916 (2022) 165219.

H. Kohlmann, Hydrogen order in hydrides of Laves phases, *Z. Krist-Cryst Mater.*, 235 (2020) 319-332.

2. YFe₂ structure and H sites

Cubic C15 structure

Fd-3m S.G. $a = 7.36 \text{ \AA}$



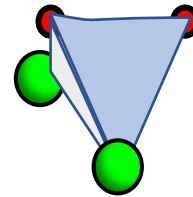
$$R_Y/R_{Fe} = 1.29 > 1.225$$

$$d_{Fe-Fe} = 2.60 \text{ \AA} > 2.49 \text{ \AA} (\alpha\text{-Fe})$$
$$d_{Fe-Y} = 3.05 \text{ \AA}$$
$$d_{Y-Y} = 3.186 \text{ \AA} < 3.56 \text{ \AA} (Y)$$

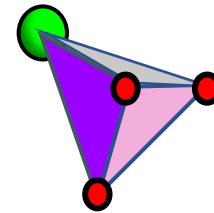
3 tetrahedral sites for H insertion

Y 8b

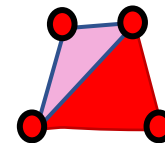
Fe 16c



Y₂Fe₂ 96g : 12 H/f.u.

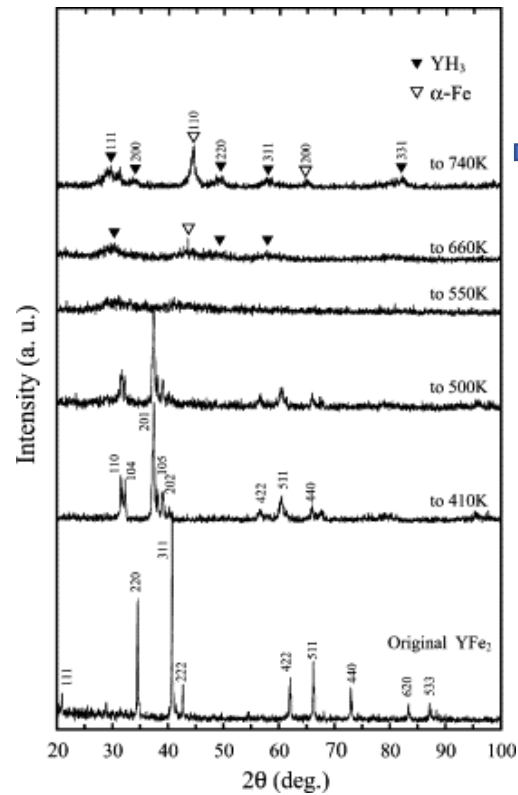


YFe₃ 32e : 4 H/f.u.



Fe₄ 8b : 1 H/f.u.

2. Hydride synthesis: Influence of T and P ?

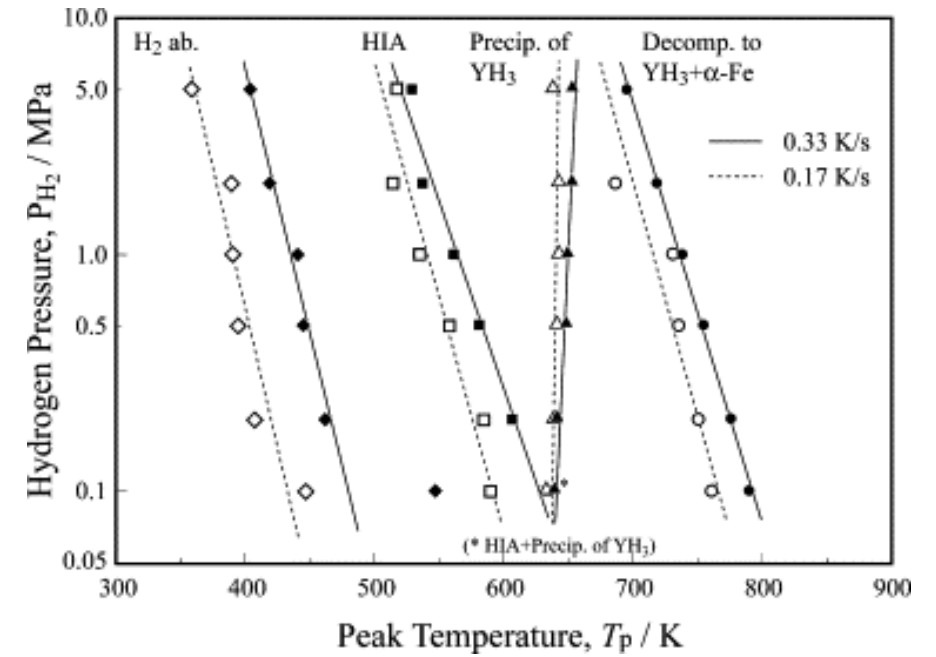


Decomposition in Fe and YH_3

Amorphisation (HIA)

Cristalline hydride

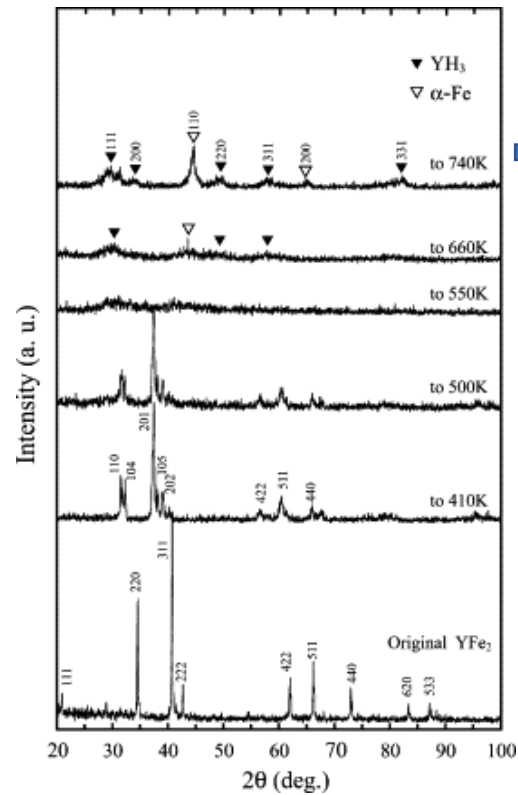
XRD patterns of YFe_2 heated a rate of 0.17 K/s and at the H_2 pressure of 1.0 MPa.



DSC: Peak temperature, T_p , of each reaction and the hydrogen pressure at two heating rates (10 and 20 K/min)

K. Aoki, H. W. Li, M. Dilixiati, and K. Ishikawa, Mater. Sci. Eng. A 449-451, 2 (2007).

2. Hydride synthesis: Influence of T and P ?

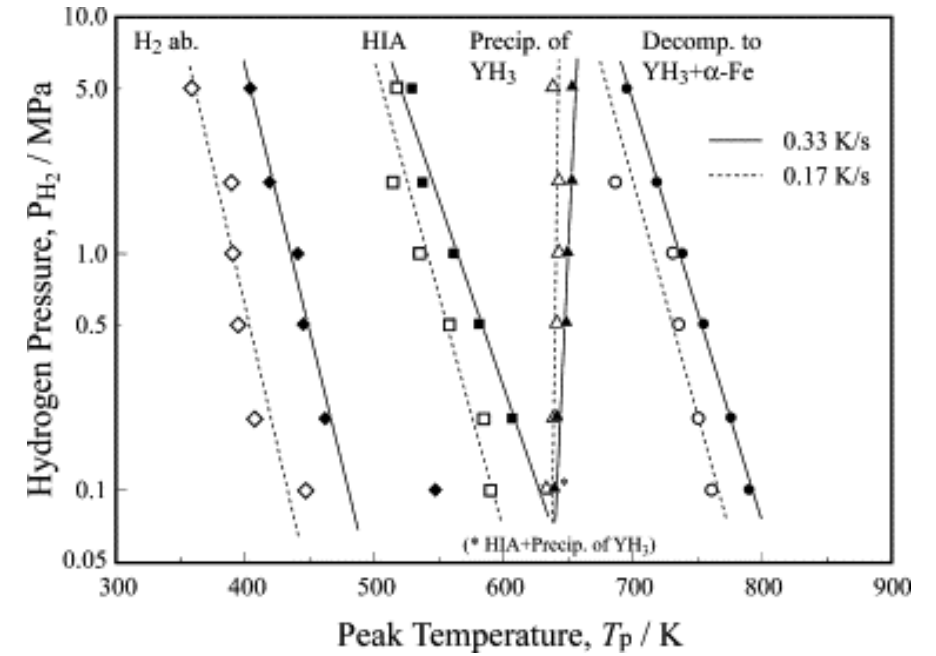


Decomposition in Fe and YH_3

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2. Questions ?

How much hydrogen can be inserted in YFe_2 ?

In which sites?

What is the influence on H insertion in YFe_2 on:

- Thermodynamic
- Structural
- Magnetic



Properties ?



3. Synthesis methods

Alloy synthesis



Induction melting Y+Fe

+3 weeks annealing
at 1100 K

In silica tube under vacuum

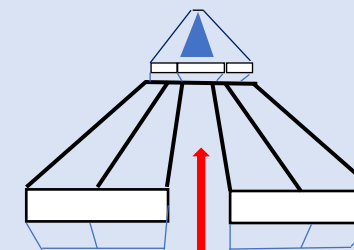
Hydride/deuteride synthesis



Home made
Sievert apparatus
 $P < 10 \text{ MPa}$, 410 K



High pressure
apparatus
 $P < 1 \text{ GPa}$, 373 K



SXR

Diamond anvil

H_2 pressure
 $P < 100 \text{ GPa}$



S.M. Filipek, Poland

P. Loubeyre, CEA-DAM

3. Characterization methods

Alloy composition



EPMA

Structure



XRD laboratory

Transition temperatures



DSC, Q100 TA instrument

Large scale facilities (structures)



Neutron center



Synchrotron center



Beam line



Magnetic measurements



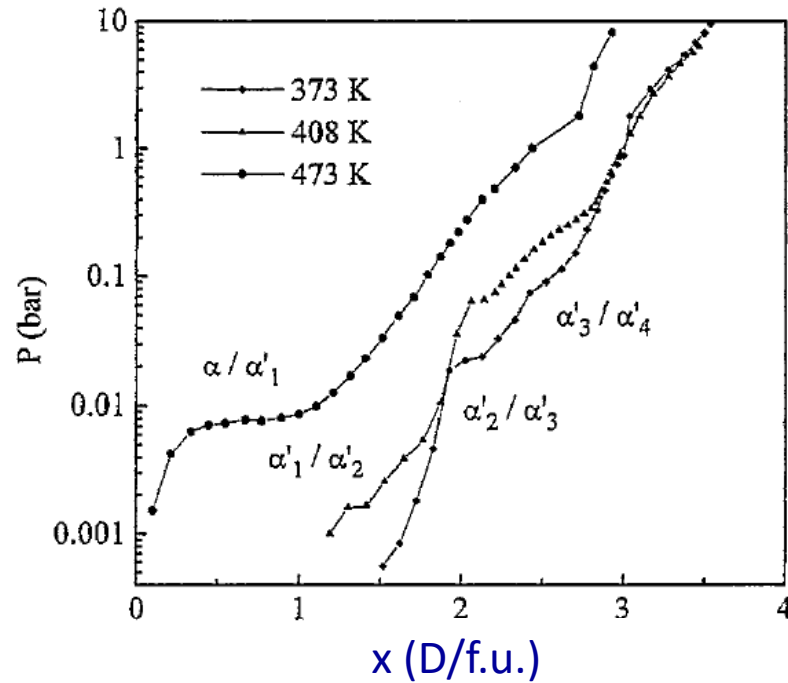
PPMS (QD),
9 T, 2-330 K



MANICS DSM8,
1.7 T, 300-900 K

4. Results

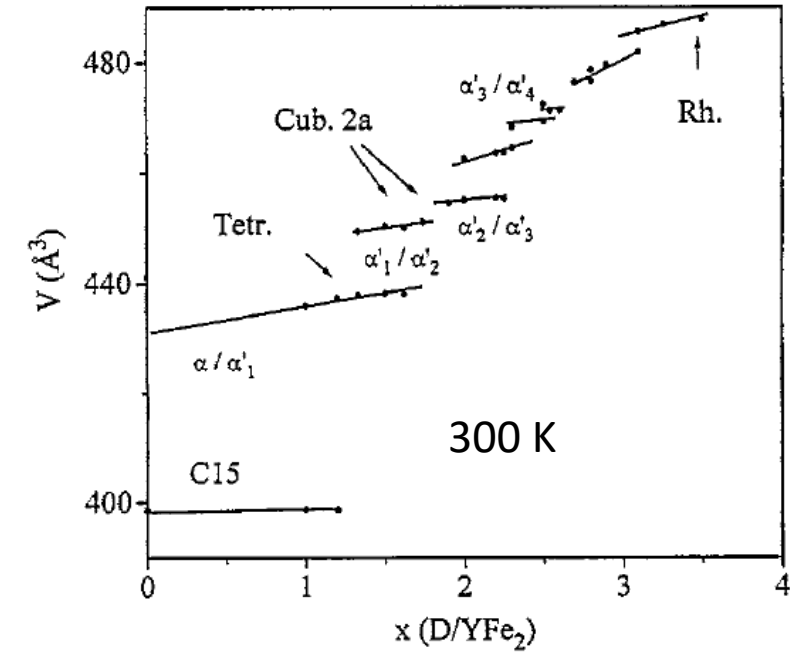
Multiplateau isotherm



Several plateaus in PCT curves



Multi-phases

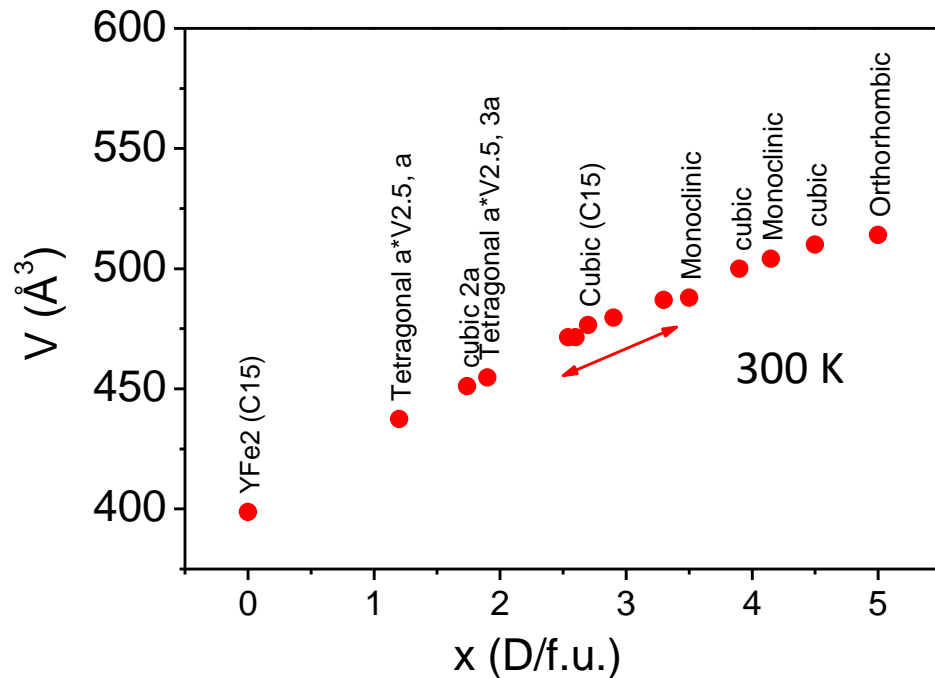


Several hydrides with different structures separated by two phase ranges

V. Paul-Boncour, M. Latroche et al, J. alloys Componds 255, 195-202 (1997).

4.1 Structural properties

SXRD + Neutrons : structure of single phase hydrides



- Cell volume increases versus D
- Lowering of crystal symmetry

Maximum H content?

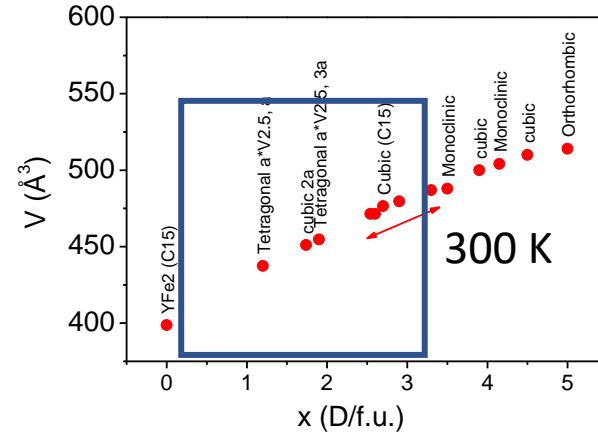
If all sites were filled 17 H/f.u.
But H-H > 2 Å and Fe₄ site is empty

The capacity depends on the applied pressure:

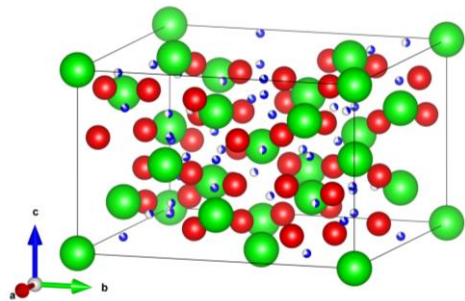
- Up to 1 MPa: **4.2 H/f.u.**
- under 0.8 GPa: **5 H/f.u.**
Remain stable when releasing pressure
- Under 35 GPa : **7-8 H/f.u.**

V. Paul-Boncour, L. Guénée, et al, J. Solid State Chem. 142, 120-129 (1999).

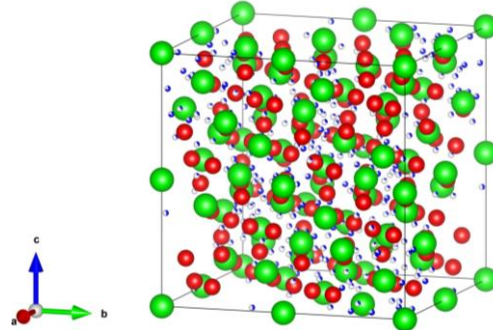
4.1 Structural properties: distortion and surstructures ($x < 3$)



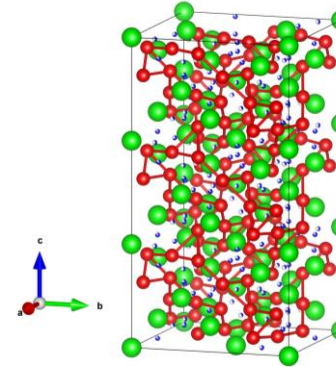
Only A2B2 sites
 Alternance of Tetragonal and
 cubic structures



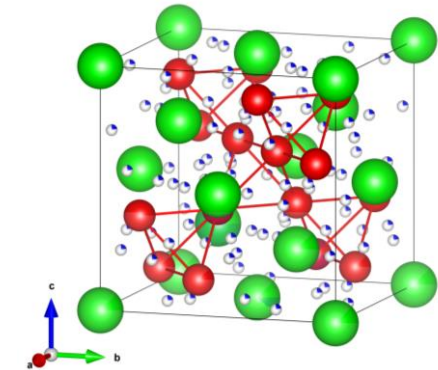
YFe₂D_{1.3} tetragonal
D in 7 A2B2 sites
 $a' = a \cdot \sqrt{5/2} = 11.985 \text{ \AA}$
 $c = a = 7.622 \text{ \AA}$



YFe₂D_{1.75} cubic 2a
D in 9 A2B2 sites
 $a' = 2a = 15.336 \text{ \AA}$

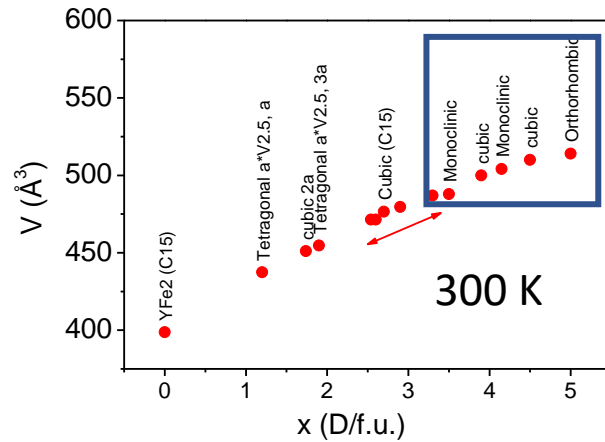


YFe₂D_{1.9} tetragonal
D in 21 A2B2 sites
 $a' = a \cdot \sqrt{5/2} = 12.15 \text{ \AA}$
 $c' = 3a = 23.079 \text{ \AA}$

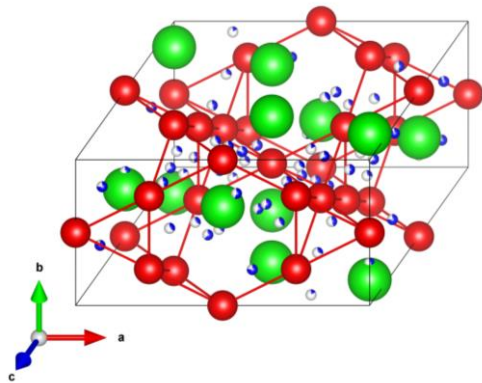


YFe₂D_{2.6} cubic a
D in 1 A2B2 sites
 $a' = 7.785 \text{ \AA}$

4.1 Structural properties: distortion and surstructures ($x > 3$)



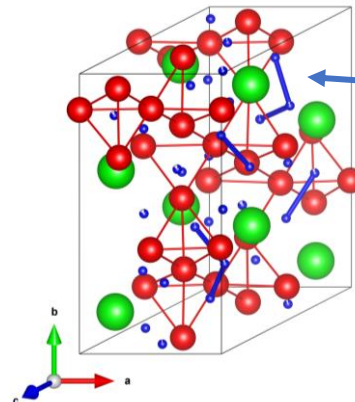
Y₂Fe₂ + YFe₃ sites
 $d_{D-D} \geq 2 \text{ \AA}$



YFe₂D_{3.5} Monoclinic (*P1c1*)
D in 9 Y₂Fe₂ + 2 YFe₃ sites

$a = 15.754(2) \text{ \AA}$, $b = 5.633(1) \text{ \AA}$
 $c = 9.476(1) \text{ \AA}$, $\beta = 144.55^\circ$

G. Wiesinger, V. Paul-Boncour, et al. J. Phys.: Condens. Matter 17, 898 (2005).

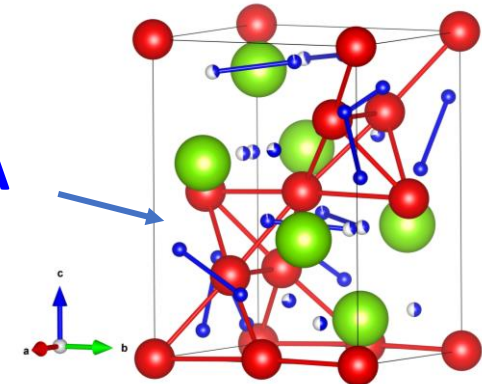


YFe₂D_{4.2} Monoclinic (*P1c1*)
D in 15 Y₂Fe₂ + 3 YFe₃ sites

$a = 5.50663(4) \text{ \AA}$, $b = 11.4823(1) \text{ \AA}$
 $c = 9.42919(6) \text{ \AA}$, $\beta = 122.3314(5)^\circ$

J. Ropka, et al., *J. Solid State Chem.* 182, 1907 (2009).

$2 < d_{D-D} < 2.1 \text{ \AA}$

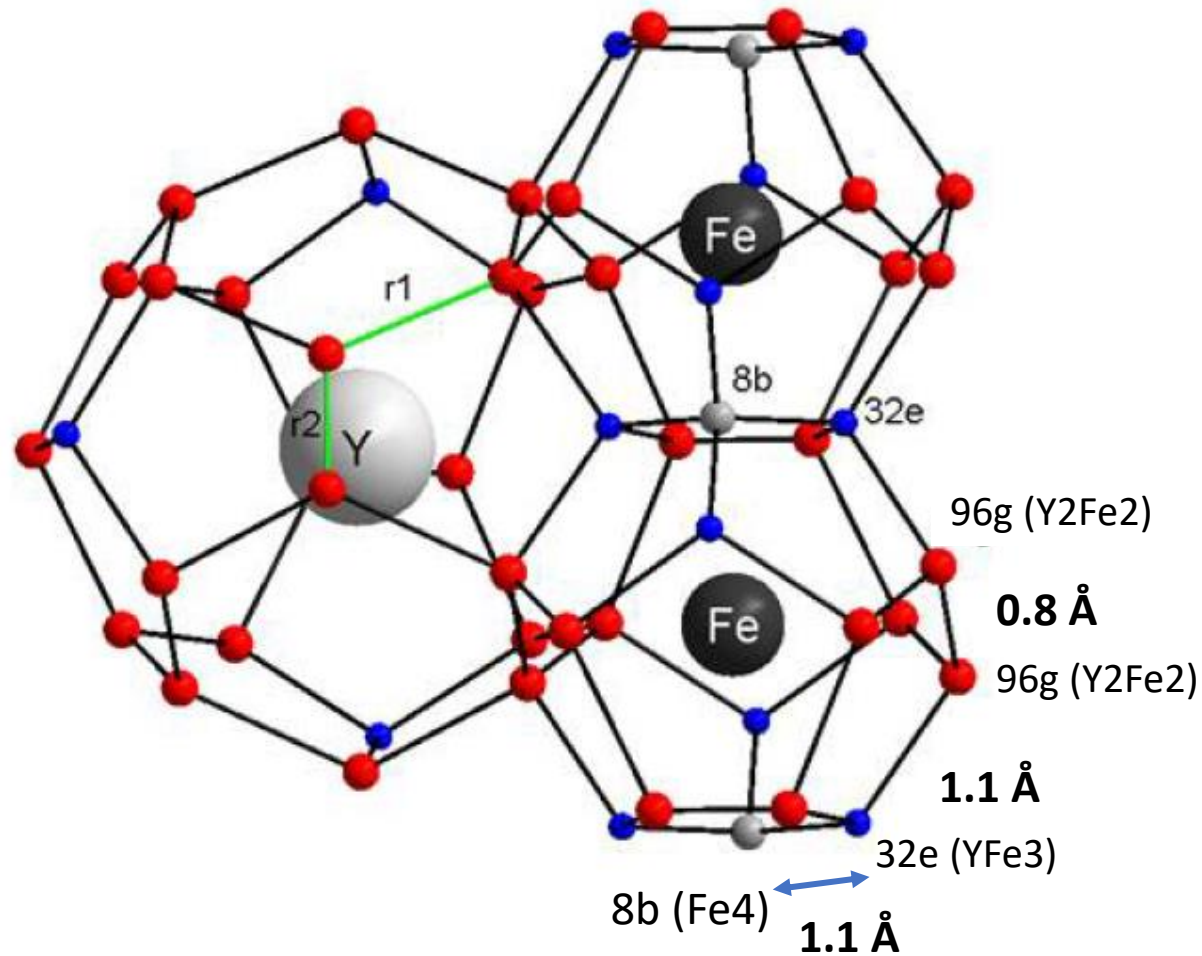


YFe₂D₅ Orthorhombic (*Pmn2*₁)
D in 6 Y₂Fe₂ + 2 YFe₃ sites

$a = 5.437 \text{ \AA}$, $b = 5.850 \text{ \AA}$, $c = 8.083 \text{ \AA}$

V. Paul-Boncour, S. M. Filipek, et al. J. Phys.: Condens. Matter 15, 4349-4359 (2003).

4. Filling of H sites



Smallest distances between 2 96 g sites= 0.8 Å

Switendick criterium:

Minimum H-H distances between H atoms > 2 Å

=> Several sites are not occupied

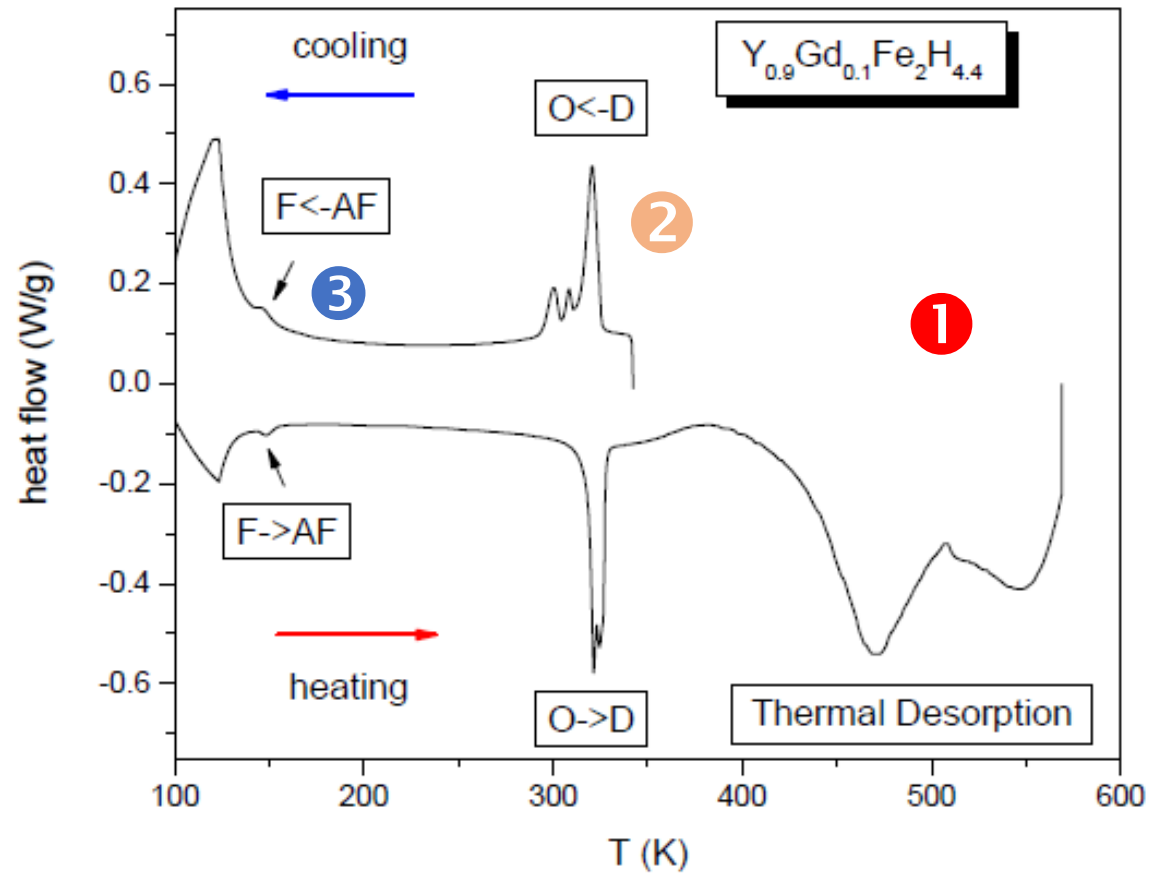
Size of the interstitial sites $r > 0.4 \text{ \AA}$

⇒ **Fe₄ site is too small to accept H atoms**

⇒ **only Y₂Fe₂ and YFe₃ sites are occupied**

4.2 Phase transitions

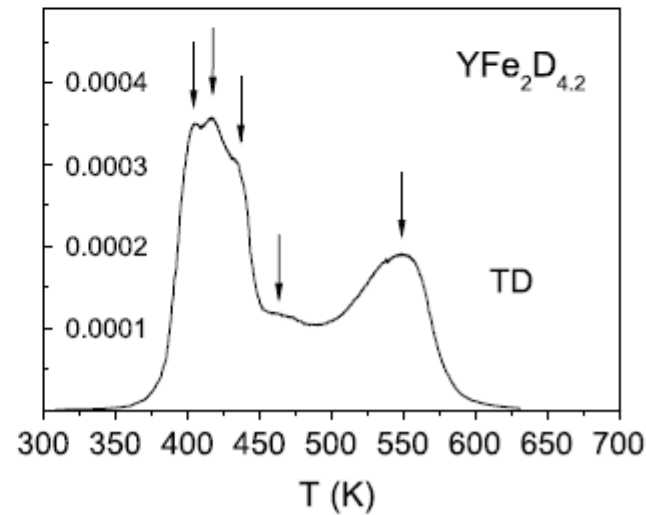
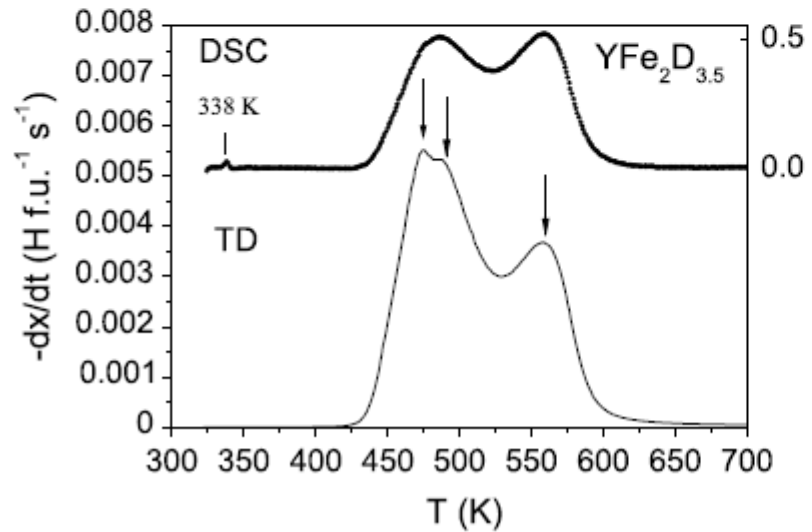
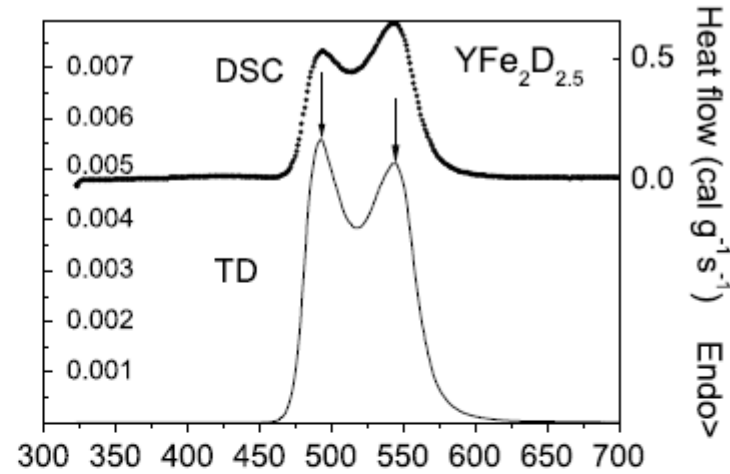
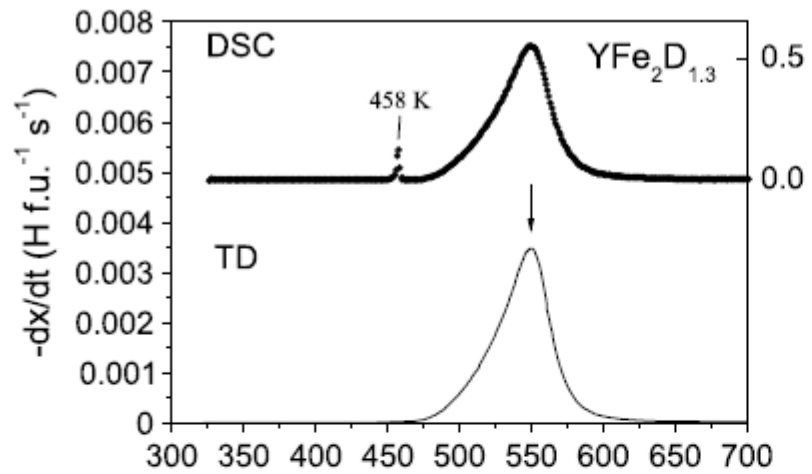
DSC



3 Different types of transition

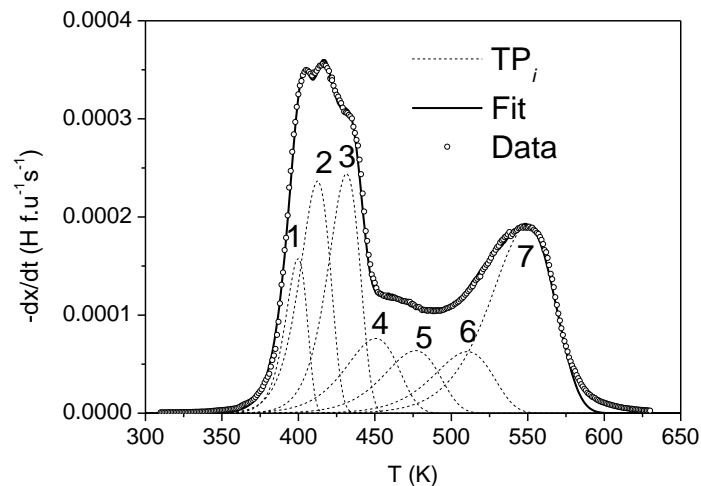
- 1 Thermal Desorption
- 2 Order-Disorder transition
- 3 Magnetic transitions

4.2.1 Thermal desorption



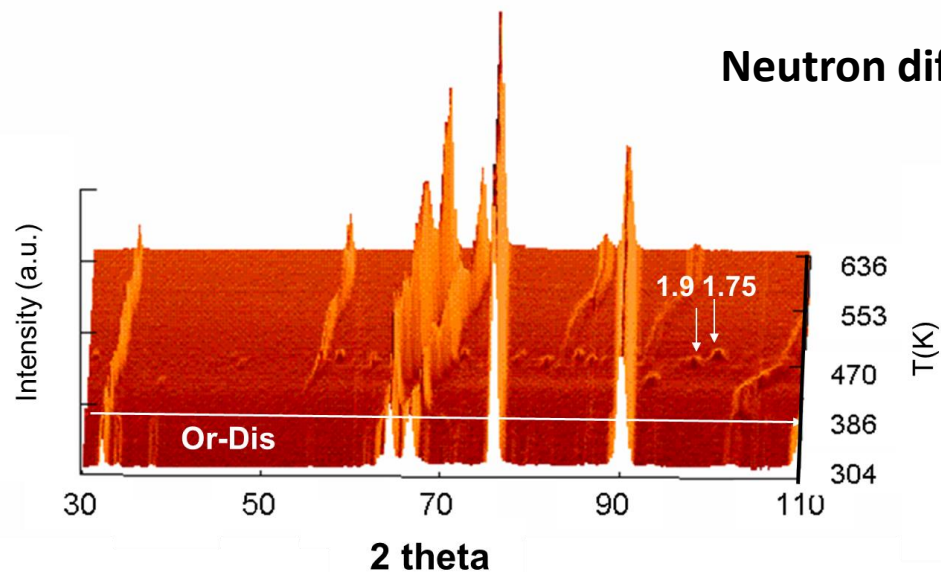
The number of peaks increases with D content: Each phase desorb at a different temperature

4.2.1 Thermal desorption $\text{YFe}_2\text{D}_{4.2}$

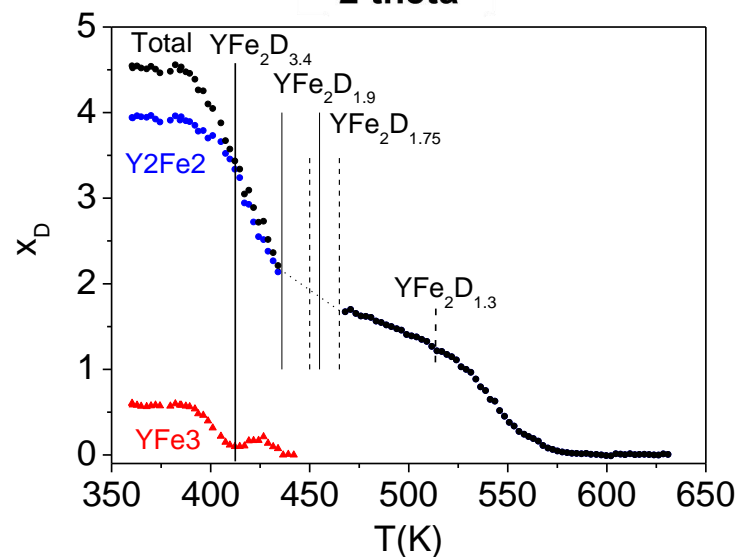


Thermal desorption (TPD):
Each peak corresponds to a phase desorption

T. Leblond, V. Paul-Boncour, F. Cuevas, O. Isnard, and J. F. Fernandez, *Int. J. Hydr. Energ.* **34**, 2278-2287 (2009).

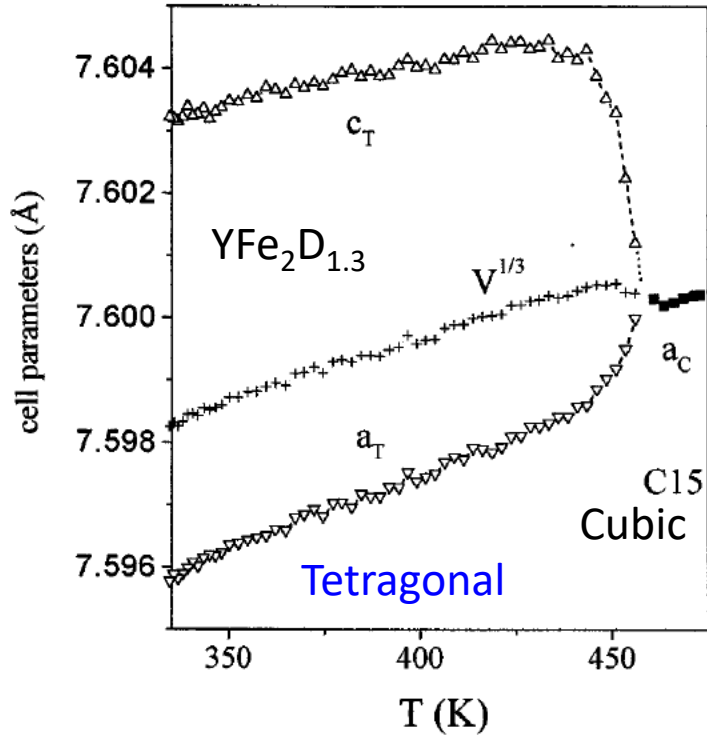


Neutron diffraction at D1B



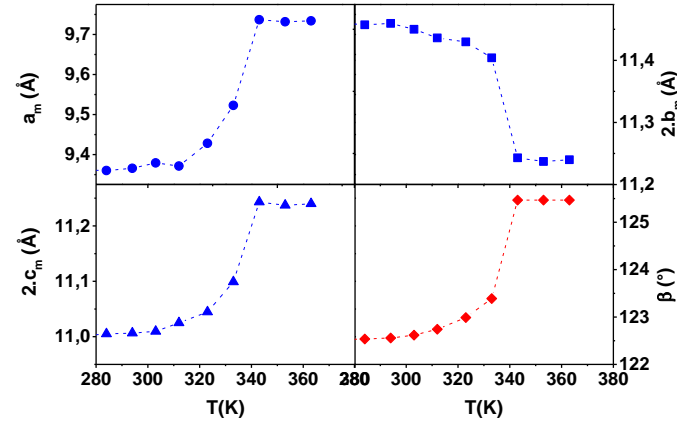
The different phases
are observed upon desorption
Confirm:
Multipeak = multiphase

4.2.2 Order-disorder transitions



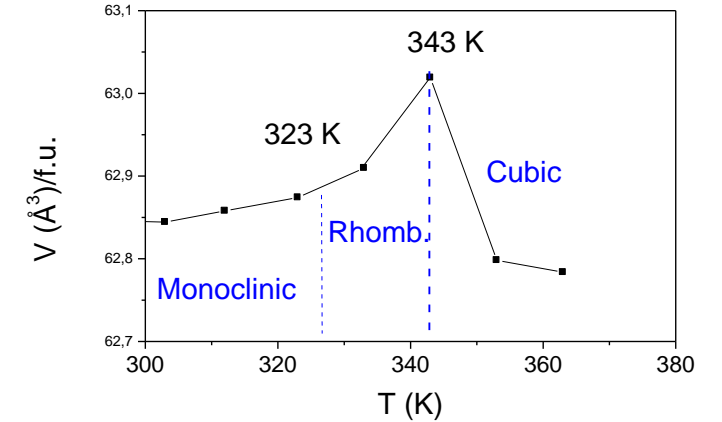
$\text{YFe}_2\text{D}_{1.3}$: At 450 K Reversible tetragonal–cubic transition

V. Paul-Boncour, L. Guénée, et al, J. Solid State Chem. **142**, 120-129 (1999).

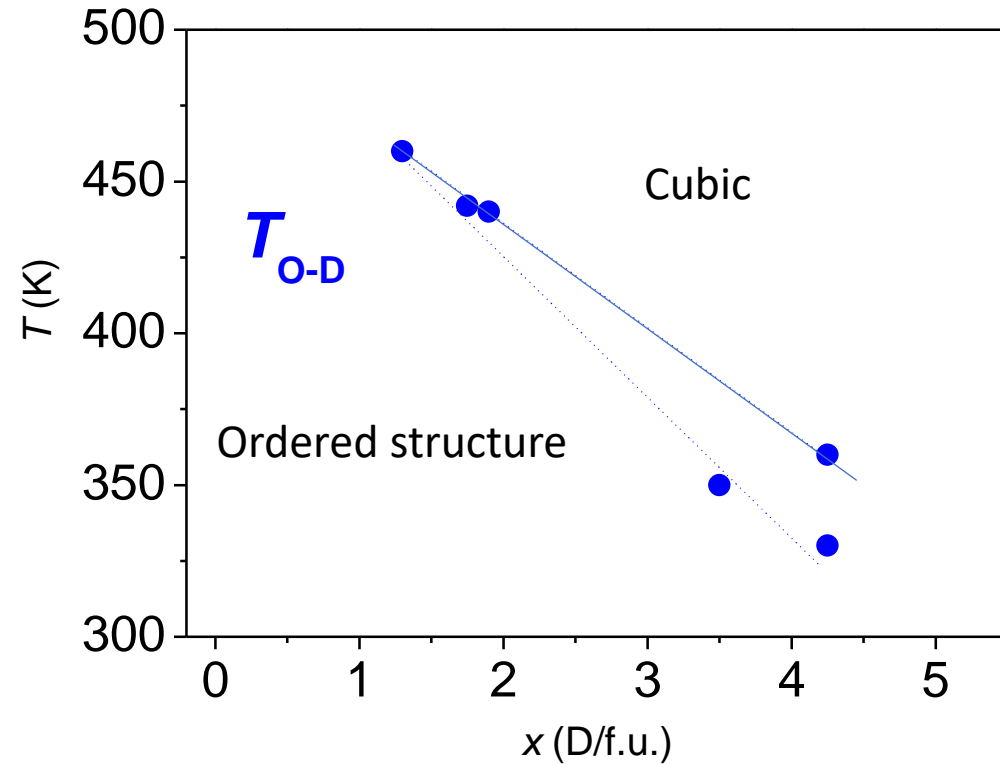


$\text{YFe}_2\text{D}_{4.2}$: 2 steps

Transition from monoclinic to cubic via a rhombohedral structure



4.2.2 order-disorder transitions



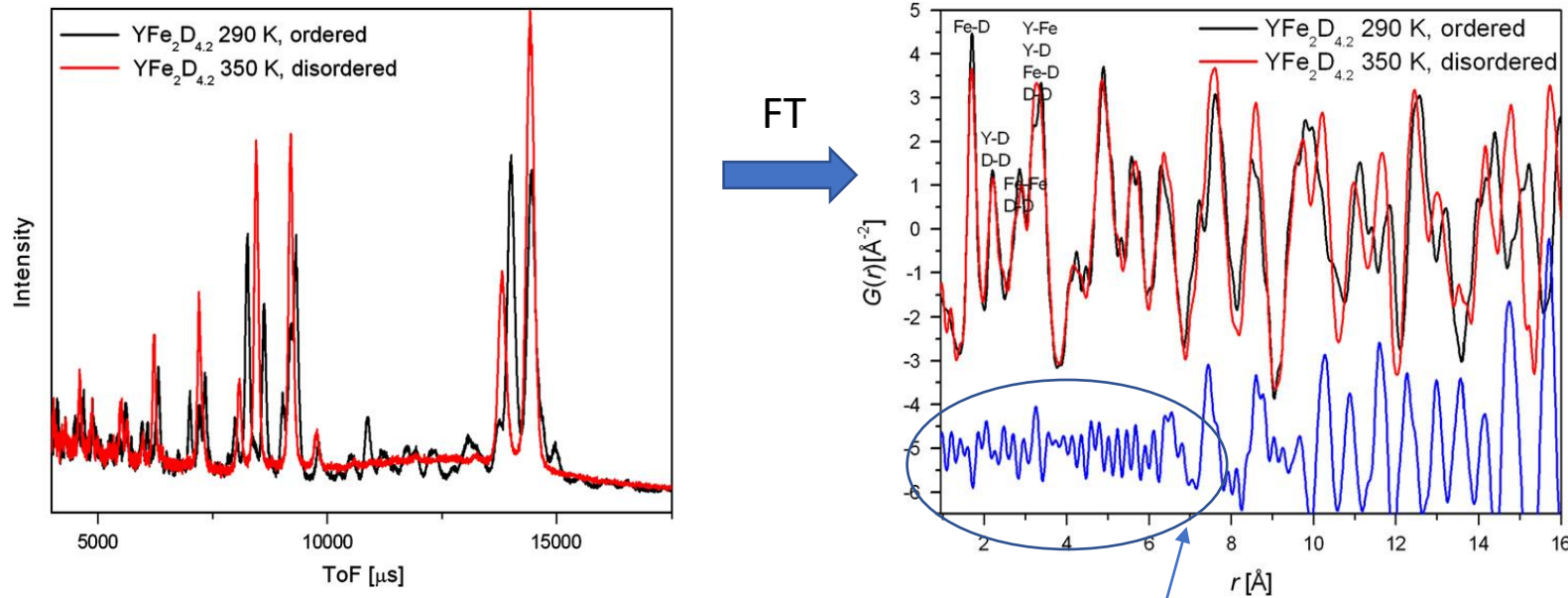
Transitions from ordered to cubic disordered phases

- T_{O-D} decreases versus D content
- $x = 4.2$: the transition occurs in 2 steps
- $x = 5$: remains orthorhombic: No O-D transition

V. Paul-Boncour, L. Guénée, et al, J. Solid State Chem. **142**, 120-129 (1999).

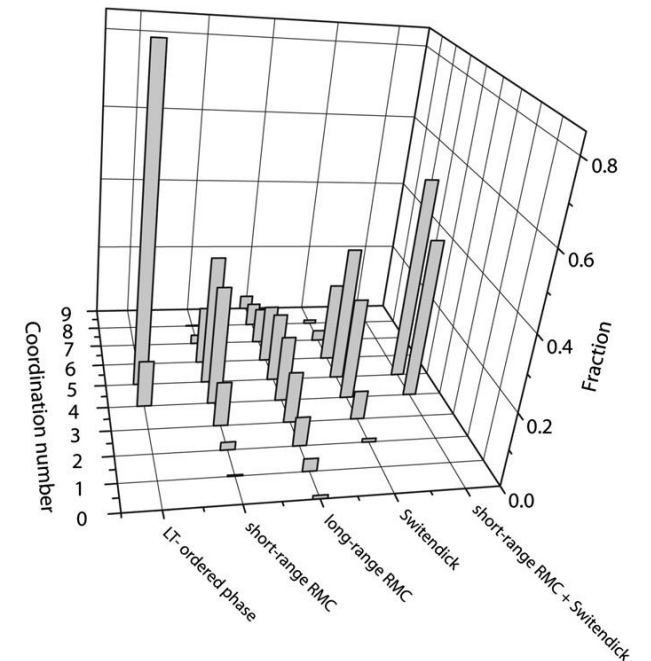
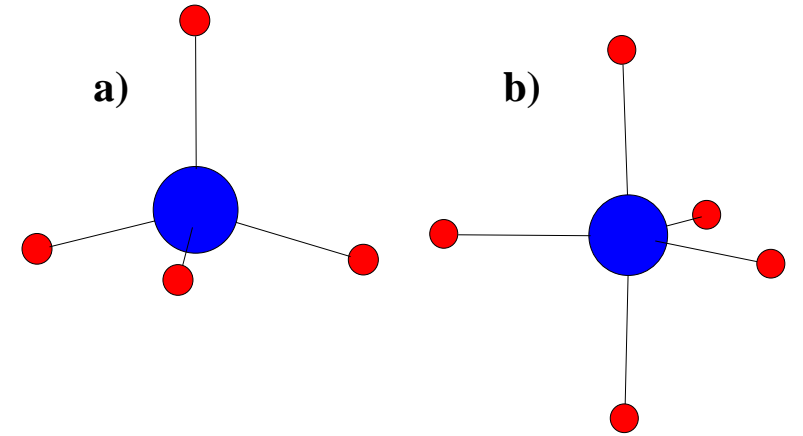
4.2.2 Order-disorder transitions

Partial distribution function = PDF (Local order)

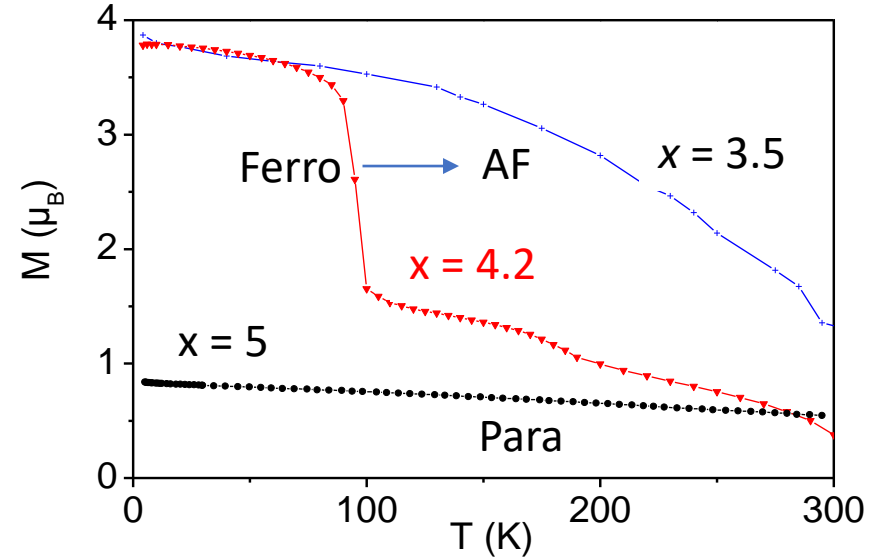
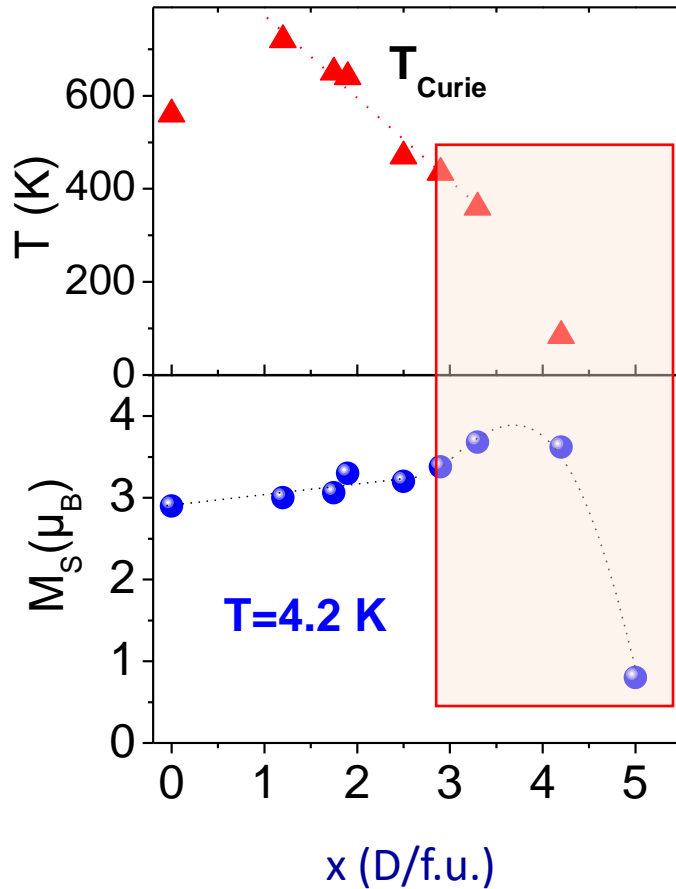


➤ The local order is preserved up to 6 Å
And diverge for larger distances (Difference)

- $d_{D-D} \geq 2 \text{ \AA}$ (switendick)
- FeD_4 and FeD_5 polyedrea (RMC): not pure random distribution



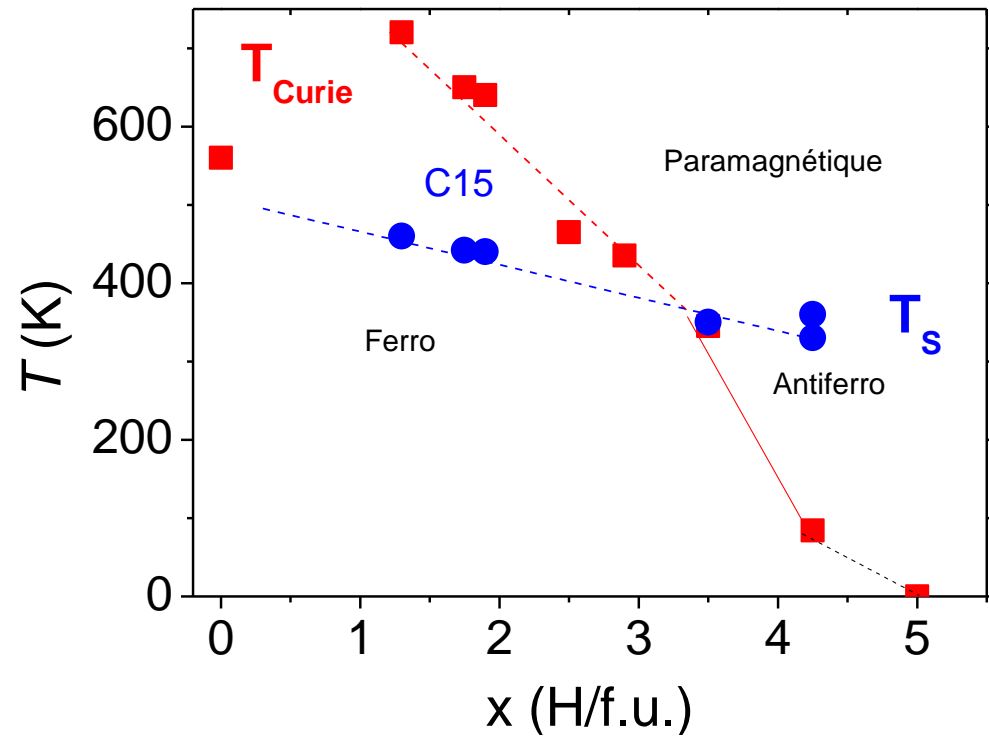
4.2.3 Magnetic transitions in YFe_2D_x compounds



Large changes of magnetic behaviour above $x= 3.5$ D/f.u.:
 $x < 3.5$: Ferromagnetic
 $x = 4.2$: Ferro-Antiferromagnetic
 $x = 5$: Paramagnetic

- Competition between cell volume increase and Fe-H bonding

4.2.3 Magnetic transitions in YFe_2D_x compounds

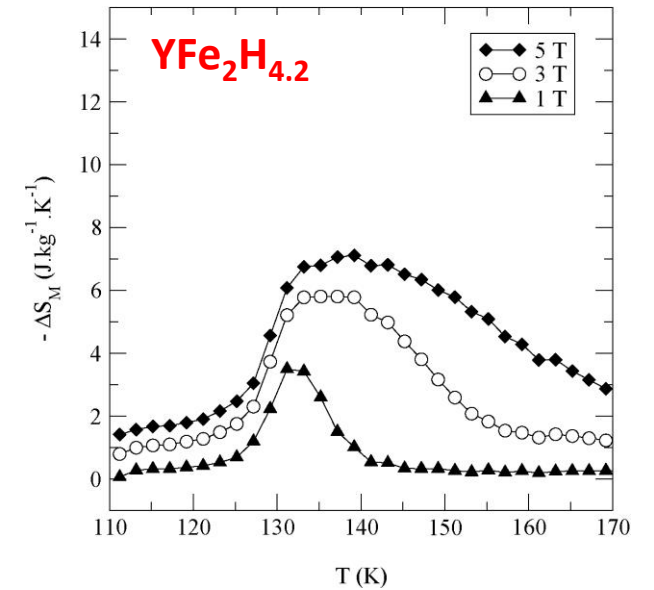
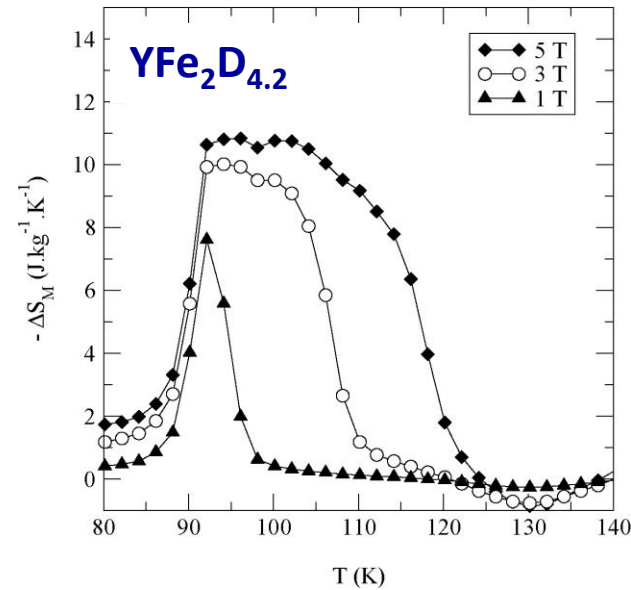
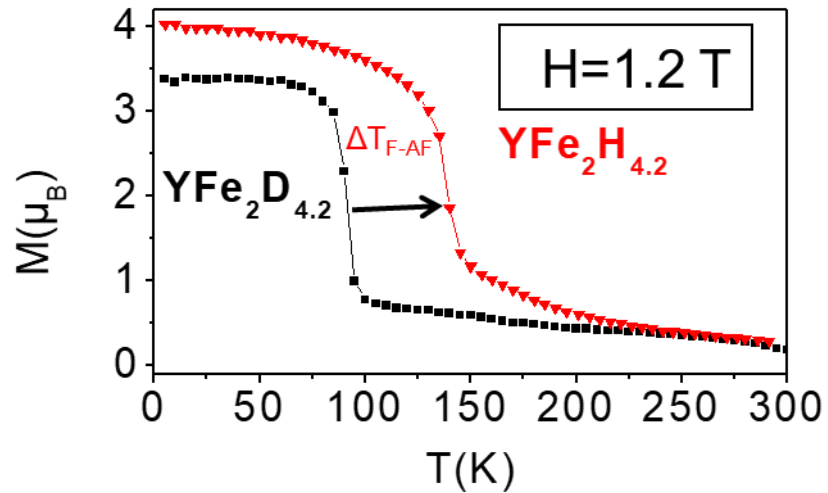


Magnetic and O-D transitions are not correlated

Larger decrease for T_{Mag} versus x

They cross at $x=3.5$

4.2.3 Isotope and magnetocaloric effect in $\text{YFe}_2(\text{H,D})_{4.2}$



Giant isotope effect:

First order ferro-Antiferro transition:

$T_{\text{F-AF}}$ increases from 84 to 131 K

$\Delta T = 47$ K or 50 % increase of $T_{\text{F-AF}}$

Magnetocaloric effect:

Large effect comparable to Gd

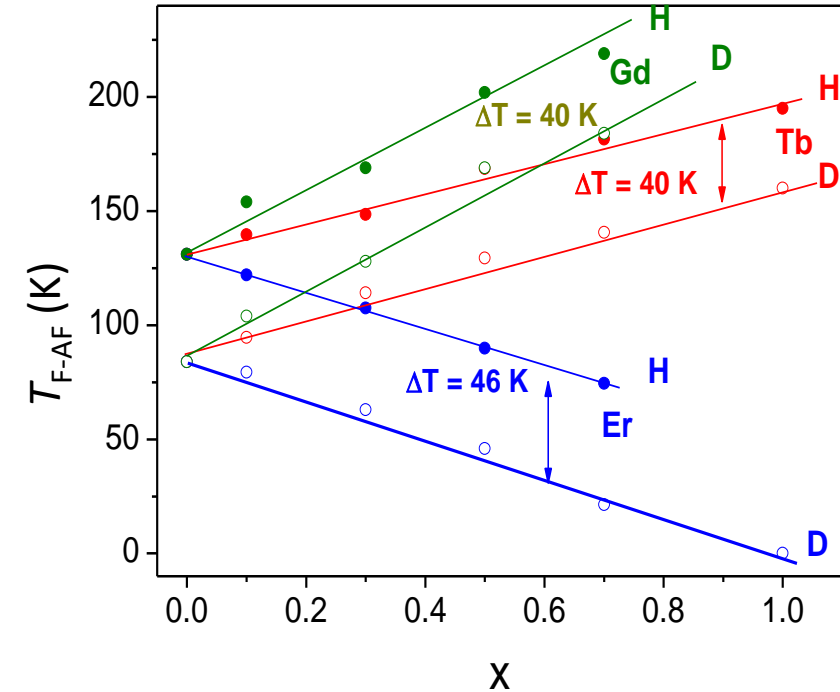
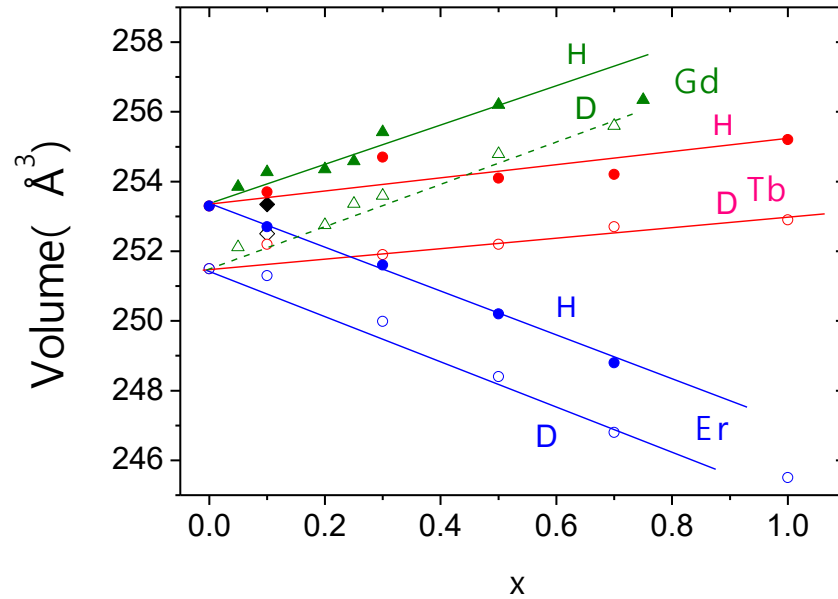
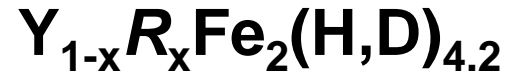
ΔS (0-5 T) = -11 and -7 J/K.kg

Magnetic refrigeration:

=> Increases $T_{\text{F-AF}}$ near room temperature

V. Paul-Boncour and T. Mazet,
J. Appl. Phys. **105**, 013914 (2009)

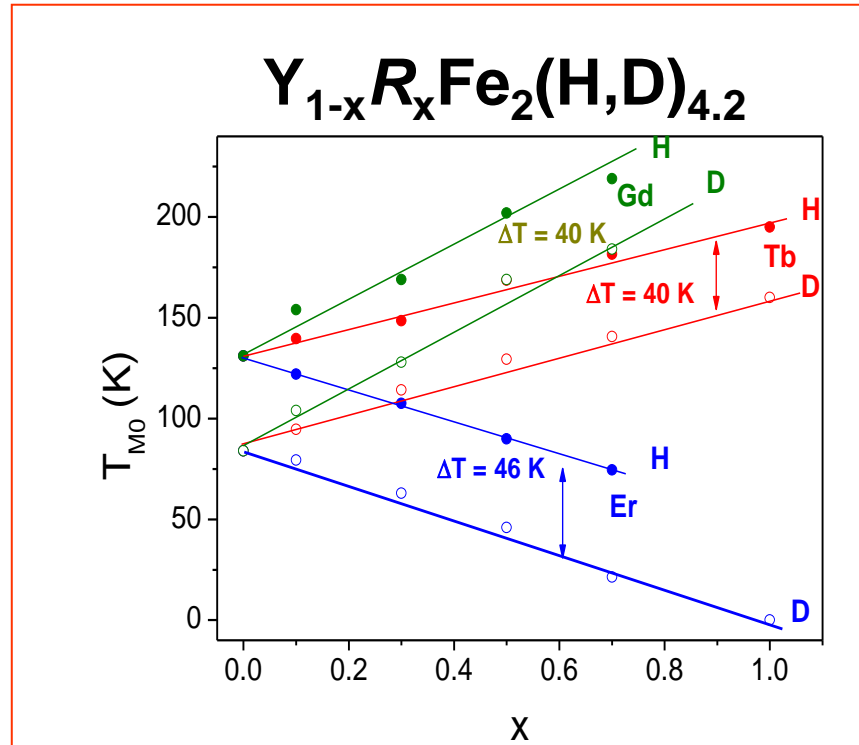
4.2.3 Influence of R for Y substitution in $Y_{1-x}R_xFe_2D_x$ compounds



To increase T_{F-AF} : change the volume by R for Y substitution

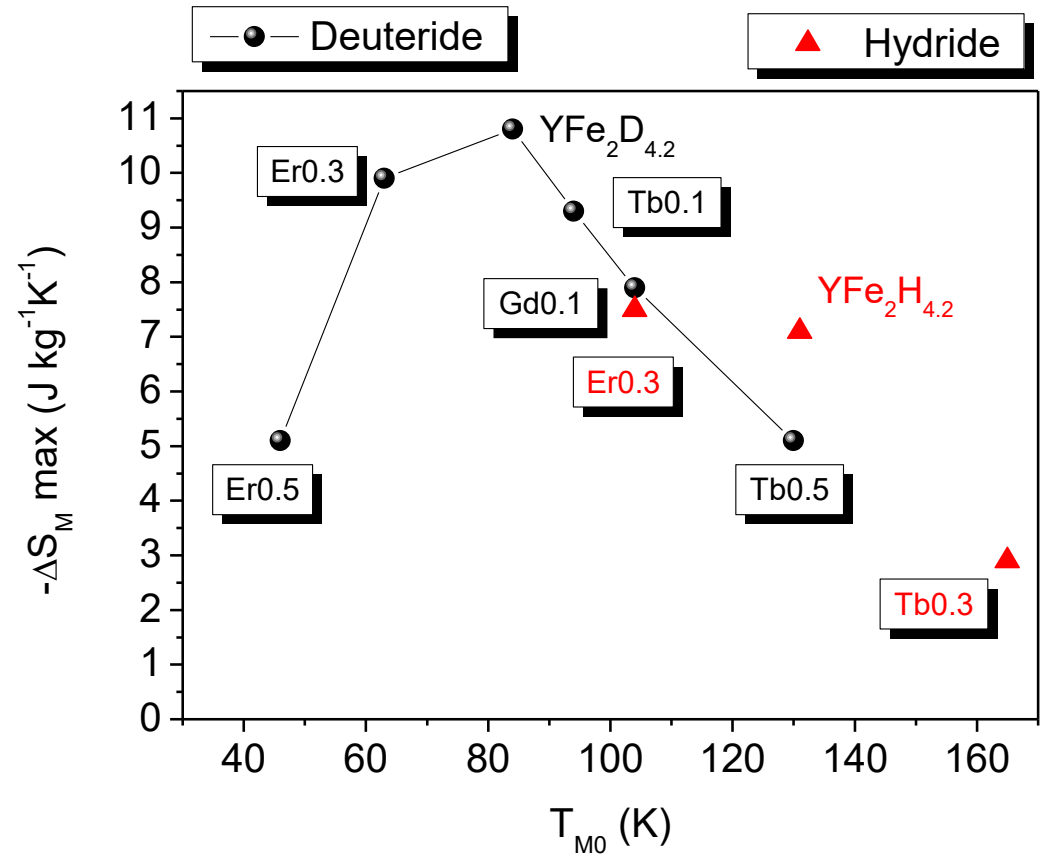
T hydride > T deuteride ($\Delta T = 40$ to 46 K)

4.2.3 Influence of R for Y substitution in $Y_{1-x}R_xFe_2D_x$ compounds



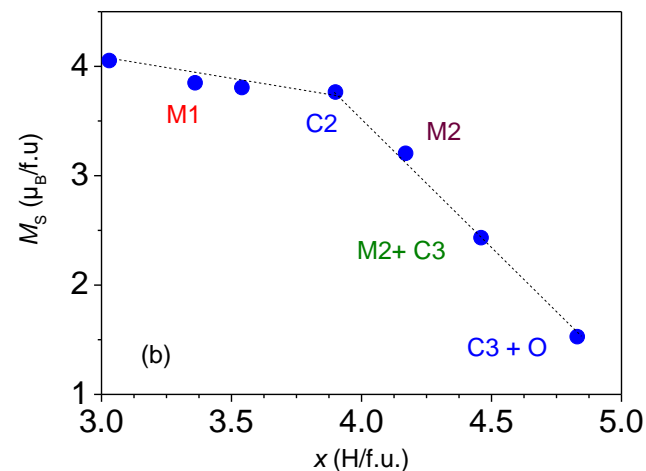
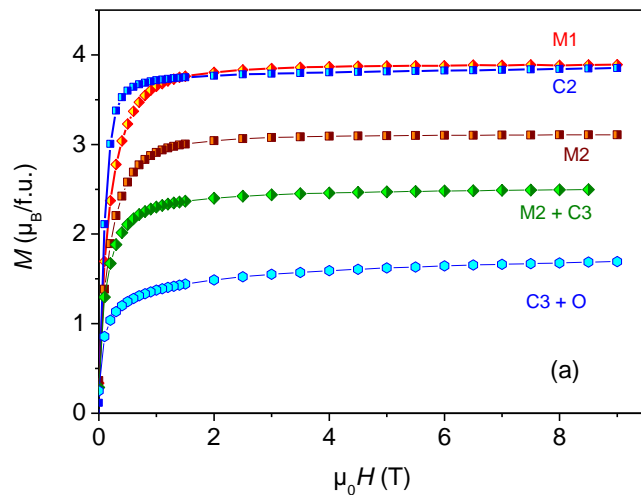
$$T_{\max} = 230 \text{ K}$$

Substitution decrease ΔS_M



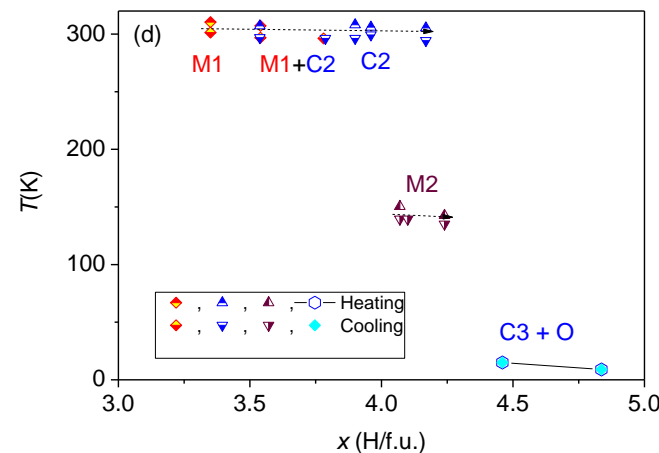
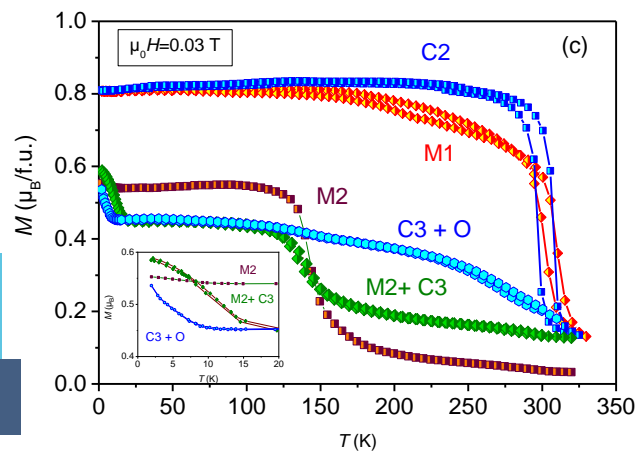
Evolution of $\Delta S_M \max$ ($\Delta H=5T$)

4.2.3 Magnetic properties of $Y_{0.9}Gd_{0.1}Fe_2H_x$



Saturation magnetization at 5 K
versus H content

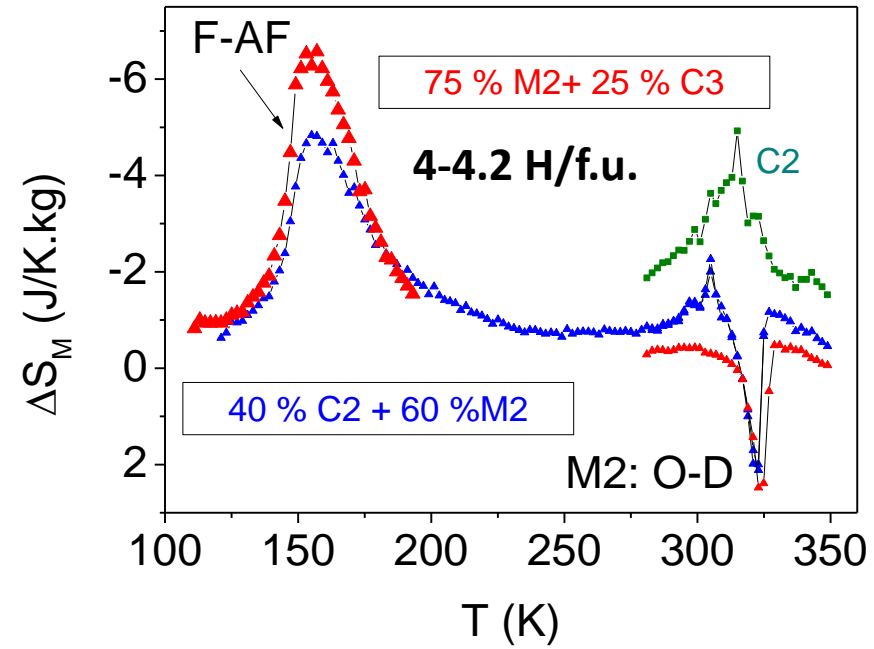
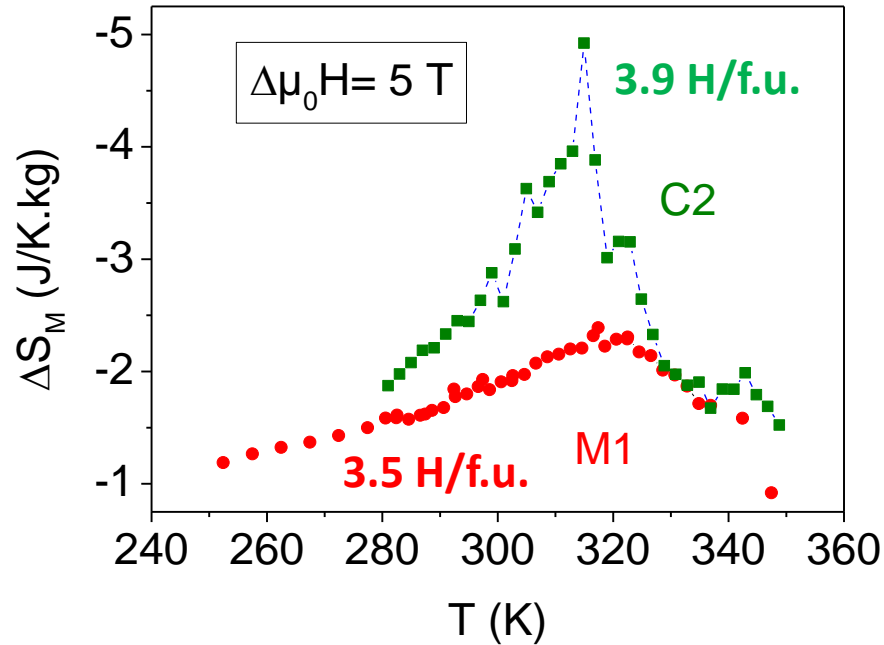
M_s decreases versus H content:
Change of dM/dx slope above 4 H/f.u.



Magnetic ordering temperatures
versus H content

T_c decreases by step versus H content:
M1 and **C2**: $T_c = 300$ K
M2: $T_{F-AF} = 150$ K

4.2.3 Magnetic entropy variations in $Y_{0.9}Gd_{0.1}Fe_2(H,D)_x$



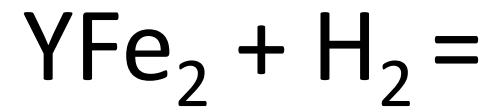
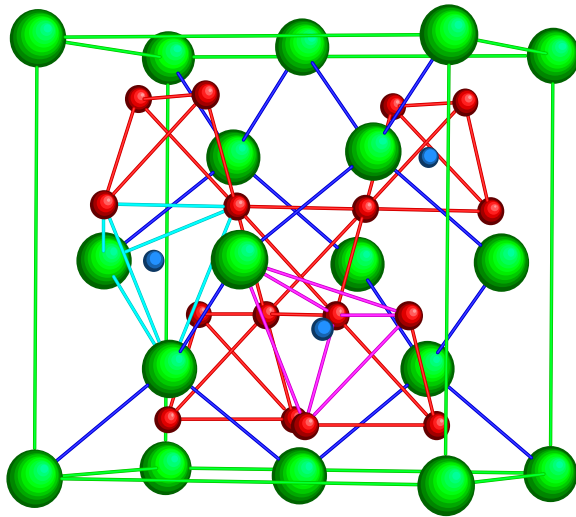
ΔS_M peaks due to magnetocaloric effect associated to magnetic transitions:

At 310K: $\Delta S_M \text{ (M1)} < \Delta S_M \text{ (C2)}$ 2nd order/ 1st order

At 150 K: $\Delta S_M \text{ (M2)}$ due to F-AF transition

At 325 K : Inverse MCE effect at 325 K due to the structural O-D transition

5. Conclusions



5. Conclusions: $\gamma\text{Fe}_2\text{H}_x$: a true Swiss knife!

Multi-states (cristalline, amorphous, nano)

Multi-plateaux isotherm

Multi-phases (> 10)

Multi-structures (> 7)

Multi-transitions

Multi-peak desorption



Multi-Magnetic entropy variation

Multi magnetic structures